

Polymeric Precursor Route for the Synthesis of Nanosized $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ Blue Pigments

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Abstract: In this work, synthesis of nanocrystalline $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ ($x=0, 0.1, 0.2, 0.4, 0.6, 0.8$ and 1) spinels by the thermal decomposition of polymeric precursors derived from metal nitrate salts and starch as a novel fuel has been systematically studied. The resulting blue pigments were characterized by the techniques X-ray diffraction (XRD), infrared (IR) and UV-Vis spectroscopy and scanning electron microscope (SEM). The XRD patterns displayed the characteristic peaks of the spinel structure and a good crystallinity. The FTIR spectra pointed out frequency bonds in range of $500\text{-}800\text{ cm}^{-1}$ correspond to metal oxygen bonds through vibrations for the spinel structure compounds. SEM images showed nano particles with less than 30 nm crystallite size. The color measurements of nano pigments were studied by using CIE L^*a^*b Parameter method which depicted the bluest color was obtained for $x=0.1$. The UV-Vis spectra of the $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ pigments confirmed the presence of tetrahedrally coordinated Co^{2+} . These facts confirmed that starch can be applied as a new and environmentally benign fuel to obtain ceramic blue pigments with $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ structure in low Co content.

Key words: Oxide materials • Nano fabrications • Ceramic blue pigment • Starch

INTRODUCTION

In recent years homogeneous and highly crystalline nano-phased metal oxide spinels have been studied by several research groups around the world for their potentials in electrochemical [1-3], magnetic [4], catalytic [5,6] and ceramic [7] applications. Generally, aluminate spinels show high thermal stability, high mechanical resistance, hydrophobicity and low surface acidity[8]. One example is CoAl_2O_4 which is well known as Thenard's blue for its impressive optical property and widely used in the ceramics, glass, paint, industry and color TV tubes as contrast - enhancing luminescent pigment[9]. However Low Co^{+2} contents are recommended since Co^{+2} is considered toxic, scarce and expensive[10]. One of the main objectives is the preparation of quality blue pigments with minimum amount of Co^{+2} under advantageous economic conditions [11]. Therefore in this work MgAl_2O_4 was chosen as a host lattice for its high melting temperature and remarkable chemical stability.

Then ceramic blue pigments from the $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ ($x=0, 0.1, 0.2, 0.4, 0.6, 0.8$ and 1) system synthesized by means of the polymeric precursor method and applying starch as an environmentally benign fuel.

MATERIALS AND METHODS

Powder Synthesis: Merck (Darmstadt, Germany) analytical reagents were used as raw materials: cobalt nitrate $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, magnesium nitrate $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, aluminum nitrate $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ and starch $\{(\text{C}_6\text{H}_{10}\text{O}_5)_n\}$. An aqueous solution containing Mg (II), Al (III) and Co(II) metal ions salts and fuel was heated at 60°C under continuous stirring after 1h, then the temperature was raised to 80°C and the mixture was stirred for several hours until a pink-reddish gel was formed. The used molar ratio $M(\text{II})/\text{Al}(\text{III})$ $\{M(\text{II})=\text{Mg and Co}\}$ was 0.5 and $M_{\text{total}}/\text{starch}$ $\{M_{\text{total}}=M(\text{II}) \text{ and } \text{Al}(\text{III})\}$ was 2.4. The blue mixed oxide obtained after a heat treatment of the metal-starch gel precursors at $T=800^\circ\text{C}$ for 1h.

Characterization Methods: X-ray diffraction patterns were recorded using a D₄-BRUKER diffractometer by Cu K α radiation at 20 KV and 30mA. A Philips XL Φ -30 scanning electron microscope (SEM) was used to observe the morphology of nano-particles. The Fourier transform-infrared (FT-IR) spectra of the gel precursors or powders were obtained on a Shimadzu FT-IR-8101 by employing potassium bromide (KBr) pellet technique. CIE-L* a* b* chromatic coordinates were determined using a Varian UV-VIS spectrophotometer (Cary 300 Bio, Mulgrave Victoria, Australia) under D65 illuminant and 10° standard observer angle.

RESULT AND DISCUSSION

The soluble starch is a low-cost, abundant, renewable, natural polysaccharide and thus fulfills environmentally aware chemistry demands [12]. Besides these facts, applying starch resulted in spinel single homogenous phase in our paper, but some other fuels like glycine leads to undesirable mixed oxides like Co₃O₄/CoAl₂O₄ in similar routs Starch has a triple role during the synthesis as complexing, template and gelation agent [13]. The starch has already been employed as a powerful tool for controlling the particle shape and size of inorganic materials and carries a large number of functional polyol groups, thereby acting as a ligand towards divalent or trivalent metal ions [14]. Moreover, at 80-90°C, the resulting complex precursors, in the presence of an excess of starch, form gels [15]. The polysaccharide acts initially as ligand toward the metal cations and determines the formation of the metals-starch complex. The helical form of the carbohydrate plays a structure-director role on the supra molecular assembly [16]. The host magnesium aluminate lattice was not chosen randomly but, rather, because MgAl₂O₄ is non-toxic/non-hazardous [17].

Fig. 1. shows the XRD patterns of the Co_x Mg_{1-x} Al₂O₄ (x = 0, 0.1, 0.2, 0.4, 0.6, 0.8 and 1) oxides heated at 800°C. For all the samples the characteristic peaks of the spinel structure were noticed, according to the ICDD files 21-1152 and 03-0896, for MgAl₂O₄ and CoAl₂O₄, respectively. The patterns did not show the presence of secondary phase. The development of the spinel phase at a relatively low temperature indicates one advantage of this synthesis route, as compared with other methods [18].

The crystallite size of the calcined pigments Co_xMg_{1-x} Al₂O₄ (x=0, 0.1, 0.2, 0.4, 0.6, 0.8 and 1) was determined based on the XRD patterns using the

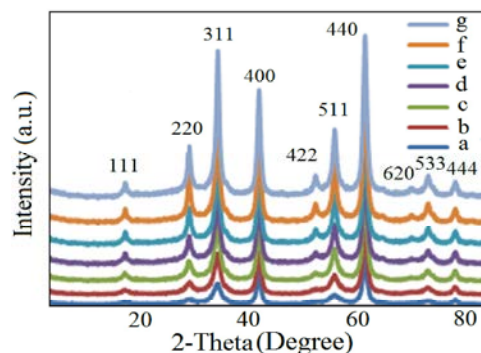


Fig. 1: Powder X-ray diffraction patterns of Co_xMg_{1-x} Al₂O₄ {x = 0(a), 0.1 (b), 0.2(c), 0.4(d), 0.6(e), 0.8(f) and 1(g)} calcined for 1 h at 800°C

Table 1: Crystallite size and colorimetric analysis of the calcined samples

Sample	Crystallite size (nm)	L*	a*	b*	ΔE
CoAl ₂ O ₄	15	37.56	-0.799	-1.023	37.58
Co _{0.8} Mg _{0.2} Al ₂ O ₄	21	38.51	-1.399	-0.046	38.54
Co _{0.6} Mg _{0.4} Al ₂ O ₄	20	39.01	-1.759	-1.211	39.07
Co _{0.4} Mg _{0.6} Al ₂ O ₄	27	39.23	-0.666	-1.485	39.27
Co _{0.2} Mg _{0.8} Al ₂ O ₄	15	42.50	-1.299	-3.676	42.68
Co _{0.1} Mg _{0.9} Al ₂ O ₄	10	67.03	-10.399	-8.796	68.39
MgAl ₂ O ₄	7	65.97	-0.168	3.455	66.06

Williamson Hall method is given in Table 1. It seems that nano pigments with mean levels of cobalt content have larger crystallite sizes while all samples have crystallite sizes below 30 nm. The chromatic coordinate (L*, a* and b*) of the pigments Co_x Mg_{1-x} Al₂O₄ (x=0, 0.1, 0.2, 0.4, 0.6, 0.8 and 1) heat treated at 800°C, are displayed in Table 1. It can be seen that the lightness, L*, decreases with the increase of the Co content, pointing out the formation of darker pigments. The coordinate a* is kept negative for all the samples, indicating a slight green condition. The b* values shows the bluest color pigment was obtained for x = 0.1 and 0.8 and the sample with x = 0.1 is the one with the darkest blue. According to Souza *et al.* higher Co-enrichment leads to darker blue pigments but the most negative values of b* observed in medium cobalt content samples in the Co_x Zn_{1-x} Al₂O₄ system [19].

The UV-Vis absorbance spectra of the samples Co_x Mg_{1-x} Al₂O₄ (x>0) displays a wide absorption in the range from 540 nm to 630 nm, corresponding to the absorption of the colors yellow, orange and red (especially for x=0.1) as present in Fig. 2. Thus, the reflectance occurs in the complementary colors, namely violet, blue and cyan, centered in the blue. The analysis of the UV-Vis spectra (Fig. 2) reveals that in the Co_{0.1} Mg_{0.9} Al₂O₄ (lowest cobalt content) three broad absorptions at approximately 540, 586 and 632 nm are well defined which are attributed to the

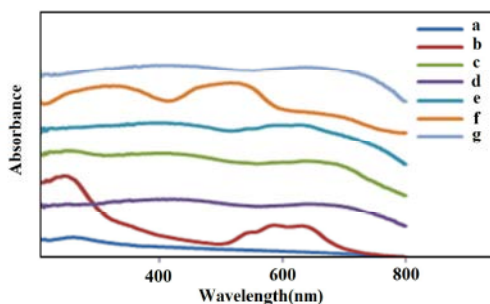


Fig. 2: UV-vis spectra of $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ $\{x = 0(a), 0.1(b), 0.2(c), 0.4(d), 0.6(e), 0.8(f) \text{ and } 1(g)\}$ calcinated for 1 h at 800°C

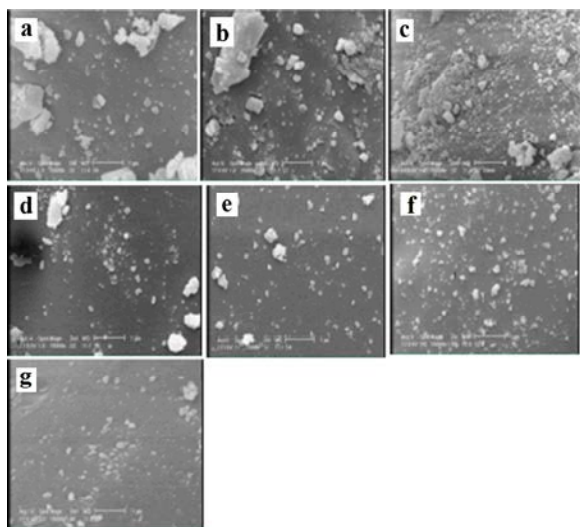


Fig. 3: SEM micrograph of the powders: $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ ($x = 1(a), 0.8(b), 0.6(c), 0.4(d), 0.2(e), 0.1(f) \text{ and } 0(g)$)

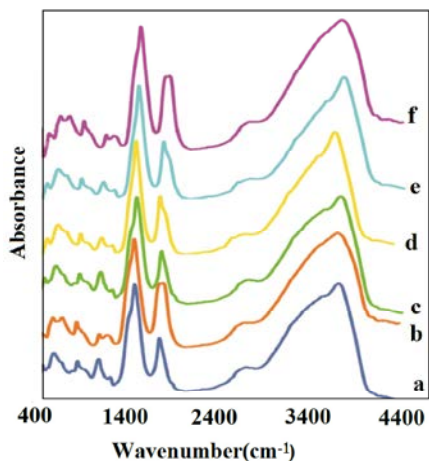


Fig. 4: FTIR spectrum for metals-starch gel precursors of $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ $\{x = 0.1(a), 0.2(b), 0.4(c), 0.6(d), 0.8(e) \text{ and } 1(f)\}$ oxides

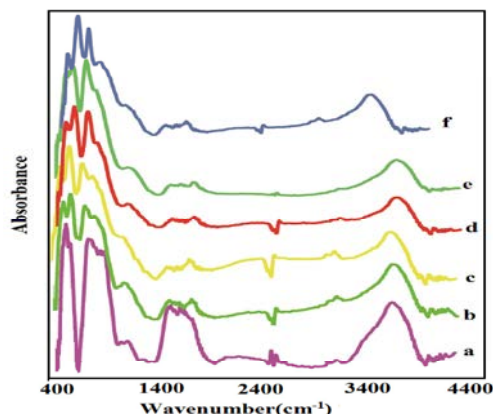


Fig. 5: FTIR spectra of $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ $\{x = 0.1(a), 0.2(b), 0.4(c), 0.6(d), 0.8(e) \text{ and } 1(f)\}$ calcinated for 1 h at 800°C

spin allowed transitions (${}^4\text{A}_2(\text{F}) \rightarrow {}^4\text{T}_2(\text{F})$, ${}^4\text{A}_2(\text{F}) \rightarrow {}^4\text{T}_1(\text{F})$ and ${}^4\text{A}_2(\text{F}) \rightarrow {}^4\text{T}_1(\text{P})$) of the Co^{+2} ions in tetrahedral sites according to the energy levels of Orgel diagram [19]. Same results were reported in the spectroscopic characterization of CoAl_2O_4 coatings by Stangar *et al.* [20] and Zayat *et al.* [21].

SEM images of the $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ ($x = 0, 0.1, 0.2, 0.4, 0.6, 0.8$ and 1) particles prepared were showed in Fig. 3. A decrease of the agglomeration degree in lower cobalt contents ($x = 0.2, 0.1$ and 0) being identified in agreement with the results of Williamson Hall method for the crystallite sizes (Table 1).

Fig. 4 and 5 exhibit the IR spectrum for gel precursor samples and resultant calcined powders of $\text{Co}_x \text{Mg}_{1-x} \text{Al}_2\text{O}_4$ ($x = 0, 0.1, 0.2, 0.4, 0.6, 0.8$ and 1) oxides, respectively. The FTIR spectra of the (Mg, Al, Co)-starch gel precursors are very similar. The stretching vibration assigned to the c-o and adjacent c-c bands overlap the $1000\text{--}1150 \text{ cm}^{-1}$ region. The result revealed a broad and strong band on the $1450\text{--}1300 \text{ cm}^{-1}$ region, which most probably results from nitrate ions. This peak disappeared in the resultant powders after calcinations at 800°C for 1h (Fig. 5). Mixed oxide spinels always display stretching bands in the $500\text{--}800 \text{ cm}^{-1}$ range, associated with the vibrations of metal-oxygen, aluminum-oxygen and metal-oxygen-aluminum [22]. These peaks are identified in gel precursors at low intensity absorption but increased in calcined powders at $T = 800^\circ\text{C}$.

In previous reports, pure and crystalline $\text{Mg}_{1-x} \text{Co}_x \text{Al}_2\text{O}_4$ powders were obtained only after annealing the as-prepared amorphous powders at high temperatures [23, 24]. Few reports on this topic show that the preparation of same ceramic pigments involves annealing

at 1400°C for 3 h [25]. Even applying new methods such as sonochemical synthesis requires heating treatment at 1000°C at least for 2 hours for the formation of pure cobalt aluminate spinel phase [26]. Generally, increasing temperature treatment increases the crystallite sizes of powders [27-29]. So preparation of single phase spinel nano particles at lower temperatures is the advantage of our liquid combustion method and makes it technically simple and cost effective.

CONCLUSIONS

Blue ceramic pigments $\text{Co}_x\text{Mg}_{1-x}\text{Al}_2\text{O}_4$ ($x=0, 0.1, 0.2, 0.4, 0.6, 0.8$ and 1) are produced as a solid solution from two mixed phases CoAl_2O_4 and MgAl_2O_4 spinels applying a starch precursor method. Powders are crystallized after calcination at 800°C and characterized by X-ray diffraction (XRD), infrared spectroscopy (IR), UV-Vis spectroscopy, diffuse reflectance spectra (CIE $L^*a^*b^*$ parameters) measurements and SEM. The crystallite sizes of spinels are estimated in the range of 7-27 nm. The bluest color was obtained for $x=0.1$ and the lightness, L^* , decreases with the increase of the Co content, pointing out the formation of darker pigments.

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